

Simulation of the local electric field in tetragonal phase of BaTiO₃

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The region of possible combinations of electronic dipole polarizability and effective charges of ions in BaTiO₃ structure at $T = 293$ K is established. It is shown that electronic polarizability of apical O²⁻ ion is anisotropic. The results obtained can be used for the computer simulations of BaTiO₃ structure peculiarities.

Установлена область возможных комбинаций электронной дипольной поляризуемости и эффективных зарядов ионов в кристаллах BaTiO₃ при $T = 293$ К. Показано, что электронная поляризуемость апикальных ионов O²⁻ анизотропна. Полученные данные можно использовать для моделирования особенностей структуры BaTiO₃.

Barium titanate (BaTiO₃) is among ferroelectric materials of the most technologic importance and being studied intensively. Both these crystals and the barium titanate based ceramics are the objects of many theoretical and experimental investigations. At high temperatures ($T > 415$ K), BaTiO₃ is a paraelectric with the cubic structure. On cooling through the transition temperature, this material becomes ferroelectric. It is believed before that the "cubic \rightarrow tetragonal" phase transition has the strongly first-order character and is associated with a remarkable optical anisotropy [1] and a large spontaneous polarization ($P_s = 0.250$ to 0.260 C/m²) [2]. Nevertheless, there are some evidences for the coexistence of order-disorder and displacement components of these phase transition in BaTiO₃ [3]. So the phase transition at 407 to 415 K in BaTiO₃ is an object of good prospects for the investigation by the computer simulation method.

Computer simulation of ionic or partially ionic systems is widely used now in solid state physics [4]. There are two approaches of its realization, namely, using either one of non-empirical (*ab initio*) methods or the classical electrostatic approach [5]. If the correct information about the effective charges of ions, their electronic polarizabil-

ity (EP) and short-range interaction parameters is available, this method is very effective. Nevertheless, the known information on this subject is inconsistent. So the main aim of this work is to determine the possible combinations of electric parameters of ions in BaTiO₃ at room temperature.

It is known that the mixed ionic-covalent type of interionic bonds results in an anisotropy of the O²⁻ ion EP, and an appreciable EP anisotropy of O²⁻ ions has been found before for KTaO₃ [6], LiNbO₃ [7] and rutile (TiO₂) [8] crystals. The lattice constants of tetragonal BaTiO₃ phase (space group $P4mm$) are $a = 3.9945$ Å, $c = 4.0335$ Å [9]. Let us denote the two O²⁻ ions localized near the center of facets as O_⊥. The O²⁻ ion localized at the 4-th order symmetry axis (apical oxygen ion) be denoted as O_∥. For the O_∥, the shortest Ti–O distance is equal to 1.9172 Å, so this bond is an ionic-covalent one. So the EP of O_∥ ion must be described by the 2-nd rank tensor with the main principal axis "3" being coincident with the 4-th order symmetry axis.

Some sets of ion EPs in tetragonal BaTiO₃ established before are listed in Table 1. It is of a principal importance that, except for α_{Ba} value, there are significant differences in the EP values of ions in the data cited. Moreover, these data are in contradic-

Table 1. Sets of EP of ions (in 10^{-24} cm^3) in tetragonal phase of BaTiO_3

α_{Ba}	α_{Ti}	α_{O}	$(\alpha_{\text{OII}})_{11}$	$(\alpha_{\text{OII}})_{33}$	Reference
1.940	0.186	2.390	2.390	2.390	[10]
1.946	0.1858	2.3928	2.3321	2.0466	[11]
1.850	0.2715	3.966	1.677	1.677	[12]

tion to the ionic-covalent binding of Ti^{4+} and O_{\parallel} ion. So, an additional independent recalculation of EP values of ions in BaTiO_3 is necessary to use those, e.g., in calculations of local electric field. To that end, the EP calculation method making use of modified Lorenz-Lorentz equation [7] was employed. This equation can be written as

$$\frac{n_k^2 - 1}{n_k^2 + 2} = \frac{1}{3\epsilon_0} \sum_{i=1}^S N_i (\alpha_{eff})_{ik}, \quad (1)$$

where n_k is the crystal refractive index for the electric component of light wave coincident with the k direction ($\mathbf{k} = \mathbf{x}, \mathbf{y}, \mathbf{z}$); S , the number of structurally non-equivalent ions in the crystal; N_i , volume concentration of the i -th ion kind; $(\alpha_{eff})_{ik}$, the so-called effective EP of i -th type of ion [7] influenced by the nearest (10 nm) dipole surrounding of these ions.

Table 2. Calculated parameters of local electric field at Ba^{2+} ion according some sets of q_{eff} and EPs of ions in BaTiO_3 at $T = 293 \text{ K}$

parameter	[10]	[11]	[12]	[9]	a	b
$q_{\text{Ba}}, e $	–	1.480	2.000	2.000	1.700	1.800
$q_{\text{Ti}}, e $	–	2.720	2.890	2.35	2.600	2.200
$q_{\text{O}}, e $	–	-1.400	-1.630	-1.45	-1.408	-1.303
						$q_{\text{OII}}, e $
$\alpha_{\text{Ba}}, 10^{-24} \text{ cm}^3$	1.940	1.946	1.850	–	2.10	1.850
$\alpha_{\text{Ti}}, 10^{-24} \text{ cm}^3$	0.186	0.1858	0.2715	–	0.150	0.200
$\alpha_{\text{O}}, 10^{-24} \text{ cm}^3$	2.390	2.3928	3.966	–	2.300	2.400
$\alpha_{11}, 10^{-24} \text{ cm}^3$	2.390	2.3321	1.677	–	1.794	1.524
$\alpha_{33}, 10^{-24} \text{ cm}^3$	2.390	2.0466	1.677	–	2.270	2.478
$E_z(\text{Ba}), 10^{10} \text{ V/m}$	–	–	-0.174	–	-0.594	-0.708
		-0.704*	-0.611*			
$V_{zz}(\text{Ba}), 10^{20} \text{ V/m}^2$ (exp. value ± 0.061)	–	-0.214	-0.209	–	-0.057	-0.050
$P_s, \text{ C/m}^2$ (exp. value 0.255)	0.260	0.241	0.261	–	0.249	0.263
		0.235*	0.255*			

* recalculated data

To calculate correctly the EP using Eq. 1, it is necessary to use the values of the crystal refractive indices in far IR band. For this purpose, we approximate the experimental dependence of ordinary (n_o) and extraordinary (n_e) indices on wavelength [13] by Sellmeier equation:

$$n_{o,e}^2 = 1 + \frac{A_{o,e}}{1 + \lambda_{o,e}^2/\lambda^2}, \quad (2)$$

where λ is the wavelength; $\lambda_{o,e}$ and $A_{o,e}$, the fitting parameters. The following values were obtained and used in the further calculations: $(n_o)_{IR} = 2.278$ and $(n_e)_{IR} = 2.244$. The EPs of ions in BaTiO_3 were analyzed using so-named bond polarizability approximation (BPA) [7]. The calculations were provided under variation of α_{Ba} and α_{Ti} within reasonable ranges of their values. Some results of calculations are presented in Fig. 1.

Because there is a great number of possible combinations of the ion EPs, the additional analysis of all these data is necessary. To verify those, two criteria were used: (1) calculation of spontaneous polarization P_s of the crystal and (2) equivalence of calculated and experimentally obtained values of electric field gradient (EFG) V_{zz} at ^{137}Ba nuclei. According to the ^{137}Ba NMR investigation, V_{zz} value at Ba ion must be

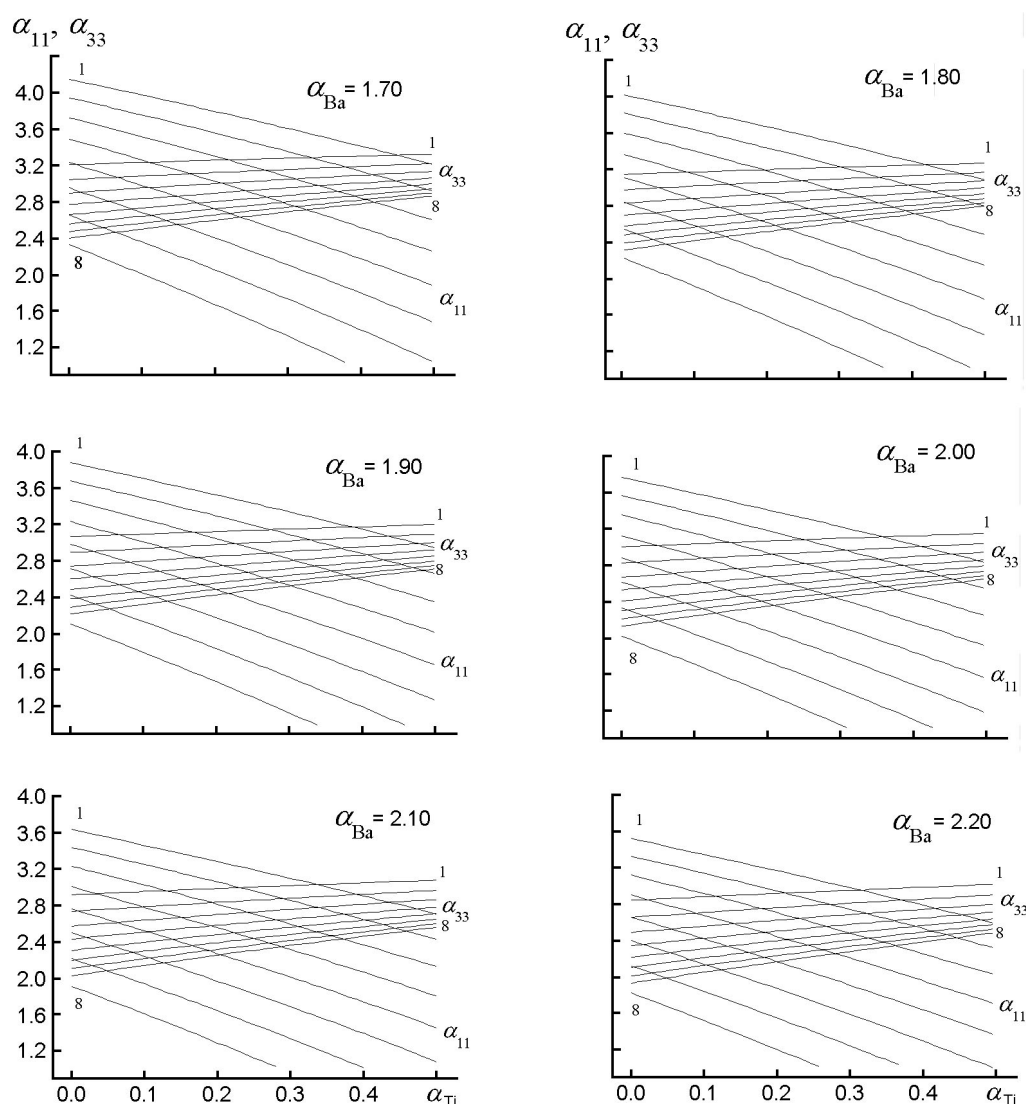


Fig. 1. Dependences of main components of a EP tensor for $O_{||}$ ions α_{11} , α_{33} for some sets of α_{Ba} and α_{Ti} values. All data in 10^{-24} cm^3 units. EP of $O_{||}$ ion varies from $1.6 \cdot 10^{-24} \text{ cm}^3$ (1) to $2.4 \cdot 10^{-24} \text{ cm}^3$ (8) with the increment $0.1 \cdot 10^{-24} \text{ cm}^3$.

$\pm 0.061 \text{ V/m}^2$ [14]. For these calculations, the additional information on the effective charges of ions in BaTiO_3 is to be used. Some sets of the known effective charges (q_{eff}) of ions in tetragonal BaTiO_3 are presented in Table 2.

The local electric field at the ions in the BaTiO_3 structure were calculated for these purposes according to the modified classical point multipole method [15]. In such way, the known sets of electrical parameters of ions in BaTiO_3 were analysed as well as new data on EPs under variation of effective charges of ions (q_{Ba} , q_{Ti} , $q_{O_{||}}$, $q_{O_{\perp}}$) within reasonable limits. Original sets of electric parameters of ions (a) and (b) contain EPs of ions according to Fig. 1 and two possible

different sets of q_{eff} . Results of the calculation of P_s , z component of E_{loc} at Ba^{2+} ion (E_z) and V_{zz} at ^{137}Ba nuclei are listed at Table 2, too. It is established that the criterion of the equivalence of calculated and experimental V_{zz} values is more strict one, in contrast to P_s analysis. Moreover, the identity of the experimental and calculated $V_{zz}(\text{Ba})$ values occur only if the relation $|q_{O_{||}}| > |q_{O_{\perp}}|$ is met. The dependences of $V_{zz}(\text{Ba})$ and P_s as functions of $\Delta q = |q_{O_{||}} - q_{O_{\perp}}|$ are presented in Fig. 2. Because $V_{zz}(\text{Ba})$ is small, it is possible that electric quadrupole moments of O^{2-} ions [16] must be additionally taken into consideration.

It is clear from comparison of (a) and (b) sets of electric parameters of ions in BaTiO_3

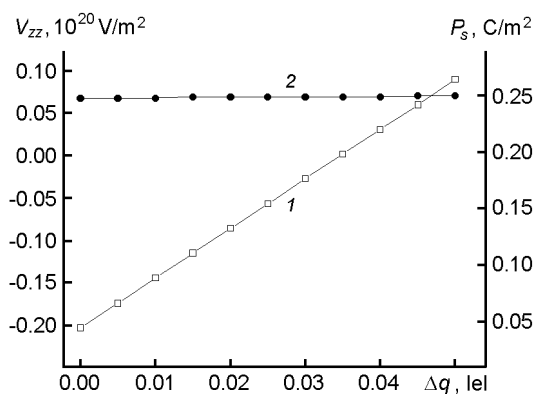


Fig. 2. Calculated V_{zz} value at ^{137}Ba nuclei (1) and P_s (2) as a function of $\Delta q = q_{\text{OI}} - q_{\text{OII}}$.

at $T = 293$ K that the additional verification of these data is necessary. A probable way to solve this problem is to analyze the balance between the long range Coulomb forces and interionic repulsive forces acting at Ba^{2+} and Ti^{4+} ions. After additional testing, these sets (or other ones estimated by us to be suitable sets of electrical parameters of ions) can be used for the computer simulations of BaTiO_3 physical properties. In any case, the both sets (a) and (b) are in a good agreement with the peculiarities of BaTiO_3 structure.

Моделювання локального електричного поля у тетрагональній фазі кристала BaTiO_3

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Встановлено область ймовірних комбінацій значень електронної дипольної поляризованості та ефективних зарядів іонів у кристалі BaTiO_3 за $T = 293$ К. Показано, що дипольна електронна поляризованість апікальних іонів O^{2-} суттєво анізотропна. Отримані дані можна використовувати для моделювання особливостей структури BaTiO_3 .

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