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Charge characteristics of the MOS structures with oxide films containing Si nanocrystals

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Abstract. The processes of charge accumulation in the MOS structures with SiO_2 films containing Si nanocrystals are investigated, depending on the conditions of their formation by pulsed laser deposition. High-frequency capacity-voltage characteristics of structures with the different thicknesses of films, sizes of Si nanocrystals, and their densities in the case of doping the films with gold and without it are measured. It is shown that the positive and negative charges are built-in, respectively, in the undoped films and those doped with gold. At the record of $C-V$ curves, the accumulation of a positive charge is observed. The value of accumulated charge is higher in thin films and in the films doped with gold. The obtained results testify the possibility of the use of pulsed laser deposition for creation of memory structures based on the charge capture by Si nanocrystals.

Keywords: silicon, quantum dot, capacity-voltage characteristics, non-volatile memory.

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1. Introduction

From the beginning of the 1990s, the intensive researches and developments of structures on the base of the nanocomposite system, silicon nanocrystal – silicon dioxide ($\text{Si NC} / \text{SiO}_2$), have been performed with the purpose to create light emitting sources in the visible region of spectrum, photodetectors, and memory devices on silicon (c-Si). This nanocomposite is silicon with the electronic system of dimension 0D (quantum dots) and is often named nanocrystalline silicon (nc-Si). Important information about the processes of capture of charge carriers, the current flow, the bands diagram of the M/nc-Si/c-Si/M structures is given by the studies of their electric properties, in particular, capacity-voltage ($C-V$) and current-voltage ($I-V$) characteristics. The results of studies of the capacity properties of structures with nanosize silicon are generalized in a review [1]. As a rule, $C-V$ characteristics with the shape similar to that for the MOS structures are observed.

Interest to the charge characteristics of these structures is related foremost to developments of devices of memory, working on the capture of charge carriers by sites (traps, NC). They possess the prospect of reduction of the tunnel oxide thickness. At the creation of the memory structures based on a floating gate with Si NC, there are the tasks of control over Si NC sizes, their dispersion, density, and the density of electronic states at

the interface of c-Si substrate and dielectric. Most works concern the structures containing porous silicon (por-Si) or layers of SiO_2 with implanted silicon.

The method of pulsed laser deposition (PLD) of films is characterized by the large technological flexibility, advantages of vacuum cleanliness, controllability of the erosive torch parameters, congruence of the process, compatibility with silicon technology, etc. It allows one to obtain SiO_2 films with quantum-size Si NC characterized by effective photoluminescence (PL) in the visible region of the spectrum at room temperature [2]. But, to our knowledge, the charge characteristics of the MOS structures based on these films are not practically studied. Purpose of the present work is to investigate the charge characteristics of M/nc-Si/p-Si/M structures with the nc-Si films obtained by PLD in dependence on conditions of their formation.

2. Experiment

The Al/nc-Si/p-Si structures containing nc-Si films on p -Si substrates ($\rho = 10 \text{ Ohm}\cdot\text{cm}$, (001)) were formed at the nc-Si film deposition by PLD in a vacuum chamber with residual gas pressure 10^{-3} Pa for the size separation of Si particles and the formation of films with a mirror surface [3].

A target was single crystalline silicon with a preliminarily deposited gold film of 100 nm in thickness

on its surface or without it. Doping by gold atoms was carried out for passivation of the unsaturated dangling bonds of silicon in the film and the formation of SiO_2 instead of SiO_x ($x < 2$). Introduction of gold resulted in the increase of the middle sizes Si NC and their dispersion [3, 4]. The YAG:Nd³⁺ laser beam (wavelength 1.06 μm , energy and duration of pulses 0.2 J and 8 ns, respectively, and frequency of their repetition 25 Hz), working in the Q-switched mode, scanned the target. The film deposition velocity lay in the range 2-20 nm/min, and the thicknesses of films were in the range 50-500 nm. With removing from the erosive torch axis, the film thickness decreased, as shown in Fig. 1. The films grown at large distances from the torch axis were characterized by the less sizes of Si NC and their higher concentration (see the insertion in Fig. 1). These regularities of formation of the SiO_x films with Si NC structure are a result of their deposition from the back low-energy flow of erosive torch particles [3]. The sputtering of aluminium films on the frontal and rearward surfaces completed the formation of MOS structures. The laser annealing provided the ohmic Al contact to *p*-Si substrate.

The time-resolved PL spectra were measured in the energy range 1.4-3.2 eV at excitation by the radiation of a nitric laser (wavelength 337 nm, and pulse duration 8 ns) with stroboscopic registration of a signal in the photon counting mode. The strobe width was 250 nm. The spectra of photoluminescence, successive in time, were measured with delay of the strobe in relation to the pulse of a laser up to maximal times of PL relaxation – about 20 μs .

Measurements of *I-V* characteristics were performed in darkness at applying the voltage changing on increasing and falling behaviors with a step of 0.1 V. The range of measured currents was $5 \times 10^{-11} \dots 10^{-2} \text{ A}$.

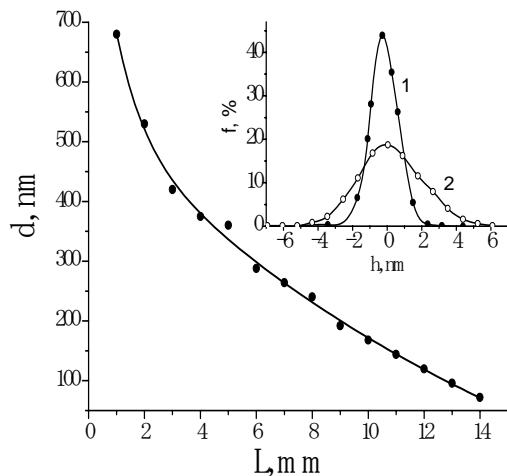


Fig. 1. Dependence of nc-Si film thickness on the distance from an erosion torch. In insert: histograms of the height distribution of surface asperities at a distance of 12 mm from the torch (1) and nearby the torch (2). Results of Atomic Force Microscopy [4].

The *C-V* characteristic measurements were fulfilled with an AMTs-1530 C meter at a testing signal frequency of 1 MHz and a voltage rate of 50 mV/s in the interval from -15 to +15 V. The measurement of *C-V* characteristics begins from the voltage corresponding to the depletion or inversion of the surface semiconductor layer (in this case of *p*-type semiconductor – from positive gate voltages) to voltages corresponding to the accumulation, and in the reverse direction. The high-frequency signal amplitude was 20 mV.

3. Results and discussion

The PL spectra of obtained films lay in the interval of radiation energies of 1.4-3.2 eV, and the times of PL relaxation lay from tens of nanoseconds to tens of microseconds. Undoped films show the PL spectra shifted to the high-energy spectral region and are characterized by rapid, less than 250 ns, PL relaxation times. The PL spectra of the films doped with gold are located in the low-energy spectral region, and the slow microsecond PL relaxation is inherent to them (Fig. 2). The resulted PL spectra of obtained nc-Si films are typical of the nanocomposite structures Si NC/SiO₂. Their photoluminescence properties, as was shown earlier in [3], are caused by radiative annihilation of excitons in quantum-size Si NC with the barrier SiO₂ layer.

I-V characteristics of the investigated structures Al/nc-Si/*p*-Si/Al had a rectifying character: the forward direction of the current corresponds to a positive potential on *p*-Si (Fig. 3). The forward branch of the *I-V* characteristics at a bias from the parts of one volt to a few volts with excluding the voltage drop on a ballast resistance corresponds to the exponential dependence. At a bias of more than 5-10 V, the saturation in the current was observed, and the current density achieved

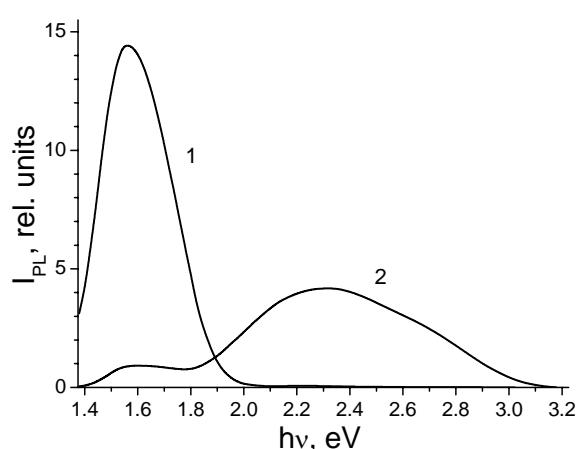


Fig. 2. Time-resolved PL spectra of the films doped with gold (1) and undoped (2).

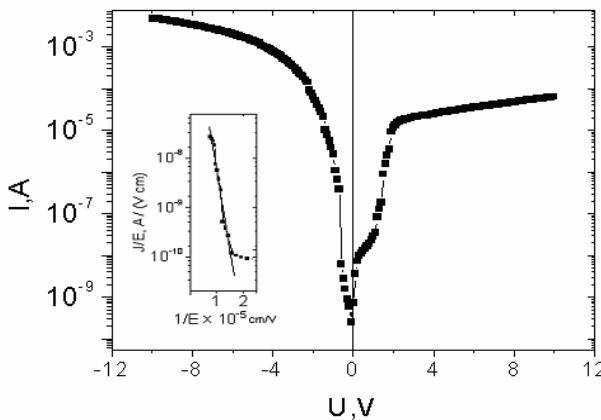


Fig. 3. Typical $I-V$ characteristics of the MIS structure with nanocomposite Si NC – SiO_2 film ($d = 173$ nm). In the insert: direct branch in the Abeles ($\ln(I/E) - 1/E$) coordinates.

the values $10^{-1} \dots 10^{-2}$ A/cm². At electric fields $E = (0.5 \dots 1) \times 10^5$ V/cm, the significant parts of $I-V$ characteristics correspond to the direct tunnelling mechanism between QD through the SiO_2 potential barriers, which is indicated by the linearity of $I-V$ characteristics in the Abeles ($\ln(I/E) - 1/E$) coordinates (see the insertion in Fig. 3) [5].

Fig. 4 shows the typical $C-V$ characteristic of the investigated structure, where the sign of voltage on the abscissa axis corresponds to the sign of bias on the M/nc-Si contact. The characteristic regions of high-frequency $C-V$ of the MOS structure are visible in the capacity-voltage (C/C_0-U) dependence. At a negative bias, the horizontal capacity line corresponding to the accumulation of holes in the space charge region (SCR) of p -Si semiconductor at the interface with nc-Si is observed. Total capacity of the structure under accumulation is determined by the capacity of the nc-Si layer, $C_{\text{nc-Si}}$, that is less in comparison with the differential capacity of SCR of p -Si, C_D . A growth of the bias in the positive direction corresponds to the subsequent realization of depletion and inversion conditions with the observed sharp drop of capacity.

Flat band voltages V_{FB} for structures with the undoped nc-Si film have small negative values (about -1 V). For structures with nc-Si films doped with gold, the V_{FB} values are about 0 V with lower dispersion.

Using the values of flat band voltages, the effective built-in charge in nc-Si films was calculated by the expression

$$Q_e = \frac{C_m}{S} (\varphi_{ms} - V_{FB}), \quad (1)$$

where φ_{ms} is the contact difference of metal (Al) and semiconductor (Si) potentials; S is the area of the metal gate electrode, and C_m is the maximal capacity of the structure.

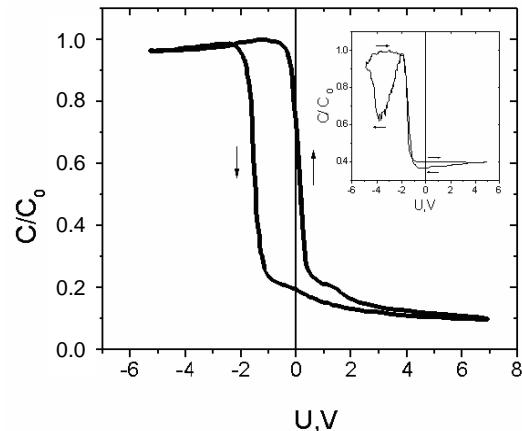


Fig. 4. $C-V$ characteristics of the MIS structure with nanocomposite Si NC – SiO_2 film. In insert: $C-V$ characteristic with the capacity decreasing under accumulation.

Calculations showed that the effective built-in charge is positive in the structures with undoped nc-Si films, whereas it is negative in the structures with films doped by gold (Fig. 5a). A positive charge is, usually, connected with E' centers (dangling silicon bonds), which arise at the oxygen deficit in a SiO_2 film, and with the oxygen vacancies which are deep hole traps [6]. As to a negative charge, the non-bridge oxygen, in particular, can be a cause ($\text{O}_3\equiv\text{Si}-\text{O}^-$ centers) [7].

Composition of the undoped nc-Si films is characterized by the SiO_x suboxides with the presence of large concentrations of oxygen vacancies [8]. As was shown in work [3], the doping of nc-Si films with gold results in the effective saturation of dangling silicon bonds on the Si NC surface and in the SiO_x matrix. The gold atoms act as catalysts for the SiO_x to the SiO_2 oxidization process. As a result, the concentration of oxygen vacancies is reduced, and the concentration of Si-O bonds is increased. These processes determine the observed signs of built-in charges.

The hysteresis loop was observed at recording $C-V$ characteristics in the forward and reverse directions. The negative sign of flat band voltage shift (ΔV_{FB}) points out to the accumulation of effective positive charge in the nc-Si film (see Fig. 4). The values of this charge calculated according to the relation

$$\Delta Q = C_m \cdot \Delta V_{FB} \quad (2)$$

as functions of the thickness of nc-Si films and the electric field strength are shown in Figs. 5b and 6, respectively. As can be seen from Fig. 5b, the value ΔQ is higher for the structures with thinner films. Higher accumulation charge in thin films can be caused by less size of NC and their higher concentration (see the insert in Fig. 1). Growth of ΔQ in the films doped with gold can be explained by a large content of negatively

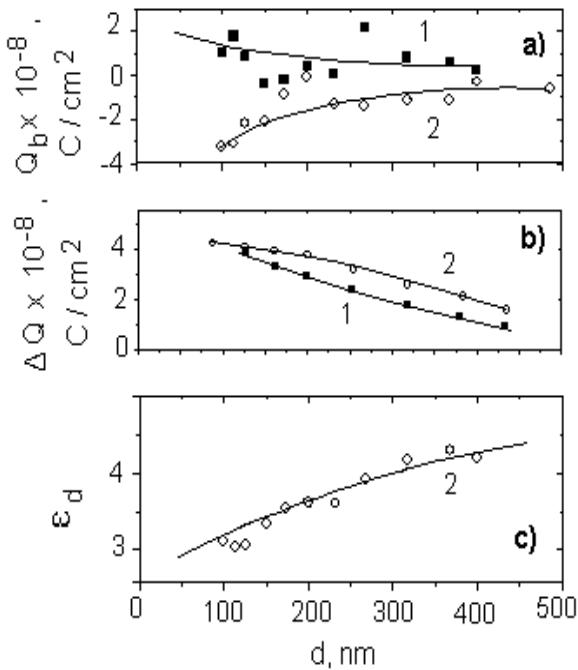


Fig. 5. Built-in charge (a), accumulated charge (b), and relative dielectric permittivity (c) vs nanocomposite film thickness: 1 – undoped film; 2 – gold-doped film.

charged defects in them caused by non-bridge oxygen. As it was expected, the increase of the accumulated positive charge with the electric field is observed (Fig. 6).

Additional measurements of $C-V$ characteristics with variation of the amplitude of a negative (positive) bias at a fixed value of the opposite-sign bias were performed. With increase in the negative bias, the considerable accumulation of positive charge took place, while only a small negative charge was accumulated with increase in the positive bias. Thus, the positive charge caused by the hole injection from p -Si SCR is accumulated in the nc-Si film. The barrier height for holes at the nc-Si / p -Si interface is high enough (about 4.6 eV) in comparison with that of the barrier for electrons. The direct tunneling of holes is complicated in this case, but the mechanism of trap-assisted resonant tunneling through traps in nc-Si is possible [9, 10]. The current under such a mechanism by orders exceeds the current of direct tunneling. The high-energy barrier for holes promotes increasing the storage time of the accumulated charge.

Reduction of capacity, sometimes with sharp oscillations, under charge accumulation in p -Si SCR is a characteristic feature of the measured $C-V$ characteristics (see the insert in Fig. 4). The Si NC charging at the increase of a negative bias causes the given effect. It is obvious that the electric field strength is sufficient for the tunneling of holes from p -Si in Si NC. At the capture of charges on Si NC, the capacity $C_{nc\text{-}Si}$ is decreased,

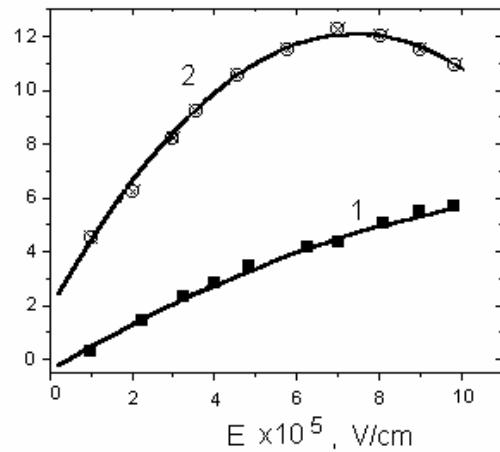


Fig. 6. Accumulated charge vs electric field strength ($d = 100$ nm): 1 – undoped film; 2 – gold-doped film.

because the number of the electronic states reactive on a small variable signal at the $C-V$ measurements is diminished. At the reverse direction of the bias, a capacity is higher than that at the direct direction, which is caused by reduction of the negative bias and, as a result, the release of holes from Si NC. The features of capacity relaxation are determined by statistical fluctuations of the potential on heterointerfaces, dispersion of the characteristic time of defects, and the Si NC recharging in nc-Si films.

Because the maximal capacity of the structure under accumulation is determined by the capacity of the nc-Si film, the relative dielectric permittivity of films (ϵ_d) can be calculated from the relation $C_m = \epsilon_d \epsilon_0 S/d$. In the case of undoped films, the thickness-related dispersion of ϵ_d is high enough. For the films doped with gold, the values of ϵ_d increase with increase in the thickness and lie in the range 3 to 5 (Fig. 5c). The large dispersion of ϵ_d in the undoped films is explained by considerable irregularities of the structure, and by a higher concentration of pores. In the strongly oxidized gold-doped films, the SiO_2 phase fills many pores, and the concentration of large Si NC is also increased with thickness. All of these determine the increase of ϵ_d with thickness.

Thus, the researches of $C-V$ characteristics of the Al/ SiO_2 film with Si NC/ p -Si/Al structures, obtained by pulsed laser deposition of the film on silicon, have shown that the structures possess the hysteresis of $C-V$ characteristics at small voltages and reveal the capture of the accumulated positive charge. It is established that the accumulation of charge is determined by the injection of holes from p -Si. Study of the influence of the formation conditions and the parameters of nc-Si films on their charge characteristics allows us to determine that the positive charge is built in the undoped nc-Si films, and the films doped with gold contain the negative charge. Higher accumulation of charge is observed in thinner

films and in the films doped with gold. The results are explained by the dependence of the capture efficiency of holes on Si NC, their sizes and density, the influence of gold on the Si NC density, and the defect composition of films. On the other hand, they testify the potential possibility for creation of non-volatile memory devices based on structures containing a dielectric with Si nanocrystals formed by the pulsed laser deposition method.

References

1. E.B. Kaganovich, E.G. Manoilov, S.V. Svechnikov, Capacity properties of the structures with low-dimensional silicon // *Optoelectronics and semiconductor techniques*, No 4, p. 61-80 (2006) (in Russian).
2. E.B. Kaganovich, S.V. Svechnikov, E.G. Manoilov, S.P. Dikiy, L.L. Fedorenko, Photoluminescence of Si nanocomposite films // Proc. VI Intern. Symp. "Thin films in electronics". Moscow-Kyiv-Kherson, Naukova dumka, Kyiv, p. 117-119 (1995) (in Russian).
3. E.B. Kaganovich, E.G. Manoilov, I.P. Bazylyuk, S.V. Svechnikov, Photoluminescence spectra of silicon nanocrystals // *Physics and Techniques of Semiconductors* **37**(3), p. 353-357 (2003) (in Russian).
4. P.M. Lytvin, O.S. Lytvin, I.P. Prokopenko, E.B. Kaganovich, I.M. Kizyak, E.G. Manoilov, Structure of photoluminescence films of nanocrystal silicon obtained by pulse laser deposition method // Proc. "Nanosystems, nanomaterials, nanotechnology" **1** (2), p. 601-610 (2004) (in Ukrainian).
5. B. Abeles, P. Sheng, M.D. Coutts, Y Azic, Structural and electrical properties of granular metal films // *Adv. Phys.* **24** (3) p. 407-461 (1975).
6. D.L. Griscom, Characterization of three E' -center variants in X- and γ -irradiated high purity a-SiO₂ // *Nucl. Instrum. Methods Phys. Res.* **B1**, p. 481-488 (1984).
7. R.A.B. Devine, The structure of SiO₂, its defects and radiation hardness // *IEEE Trans. on Nucl. Sci.* **41**, p. 125-132 (1994).
8. I.P. Lisovsky, V.G. Litovchenko, D.O. Mazunov, E.B. Kaganovich, I.M. Kizyak, E.G. Manoilov, IR spectroscopy of barrier phase of the photoluminescence nanocomposite films Si/SiO_x // *Ukr. Phys. J.* **50** (6), p. 582-585 (2005).
9. F. Compabadal, V. Milian, X. Aymerich-Humet, Trap-assisted tunneling in MIS and Schottky structures // *Phys. status solidi (a)*, **79**(1), p. 223-236 (1983).
10. J. Millan, F. Serra-Metres, X. Aymerich-Humet, Resonant tunnel switching in MIS devices // *Solid State Electronics*, **25**(7), p. 565-569 (1982).